# AI-Mg-Mn (Aluminum-Magnesium-Manganese)

V. Raghavan

The early experimental data on this system reviewed by [1995Vil] presented a partial liquidus projection and partial isothermal sections at 750, 710, 670, 500, 450, 425 and 400 °C for compositions near the Al-Mg side. Subsequently, [2005Ohn] developed a thermodynamic description of this system with emphasis on Mg-rich alloys. A liquidus projection and an isothermal section at 710 °C were presented in the update of [2007Rag] from the computed results of [2005Ohn]. Very recently, two detailed thermodynamic assessments of this ternary system by [2007Du] and [2009Shu] were published. Both assessments use descriptions of the Al-Mn system, which yield the phase equilibria in Al-rich alloys different from the currently accepted Al-Mn diagram.

## **Binary Systems**

1050

950

L

The Al-Mn phase diagram [1987Mur, 1996Liu, 2006Pre, 2007Du, 2008Gru] depicts the following intermediate phases: Al<sub>12</sub>Mn (Al<sub>12</sub>W-type cubic, denoted G), Al<sub>6</sub>Mn (Al<sub>6</sub>Mn-type orthorhombic),  $\lambda$ Al<sub>4</sub>Mn (hexagonal, space group *P*6<sub>3</sub>/*m*),  $\mu$ Al<sub>4</sub>Mn (hexagonal, *P*6<sub>3</sub>/*mmc*), Al<sub>11</sub>Mn<sub>4</sub>(HT) (Al<sub>3</sub>Mn-type orthorhombic), Al<sub>11</sub>Mn<sub>4</sub>(LT) (Al<sub>11</sub>Mn<sub>4</sub>-type triclinic), Al<sub>8</sub>Mn<sub>5</sub> (Al<sub>8</sub>Cr<sub>5</sub>-type rhombohedral),  $\gamma$  (34.5-52 at.% Mn; bcc) and  $\varepsilon$  (55-72 at.% Mn; cph). Recent

versions of the diagram [1996Liu, 2007Du] exclude the intermediate phase  $\gamma_1$  (Cu<sub>5</sub>Zn<sub>8</sub>-type cubic). Also, there is no unanimity about the nature of the phase relationships at the Al-end. The currently accepted diagram in this region [1987Mur, 2006Pre, 2008Gru] is shown in Fig. 1. Al<sub>6</sub>Mn forms peritectically at 705 °C from liquid (L) and  $\mu$ . The  $\lambda$  phase forms peritectoidally at 695 °C from Al<sub>6</sub>Mn and  $\mu$ . In their experimental investigation of this region, [2007Du] presented a different result. According to them, the  $\lambda$ -phase forms peritectically at 721 °C from L and  $\mu$ . Al<sub>6</sub>Mn also forms peritectically at 703 °C from L and  $\lambda$ . Nucleation difficulties pose problems in identifying the correct equilibria. Opinion about the relative ease or difficulty with which Al<sub>6</sub>Mn or  $\lambda$  nucleates has been divided.

[2007Du] reassessed the Al-Mn phase diagram, using their new experimental results along with the literature data. Figure 2 shows their computed diagram in full. [2009Shu] also carried out a new assessment of the Al-Mn system, using the modified quasichemical model for the liquid phase. Their computed diagram is compared with that of [2007Du] in Fig. 2. In both the diagrams, the  $\lambda$ -phase forms peritectically, which is at variance with the version in Fig. 1.

The Al-Mg phase diagram [1998Lia] has the following intermediate phases: Al<sub>3</sub>Mg<sub>2</sub> (Al<sub>3</sub>Mg<sub>2</sub>-type cubic, labeled  $\beta$ ), R or  $\epsilon$  (rhombohedral) and Al<sub>12</sub>Mg<sub>17</sub> (A12,  $\alpha$ Mn-type cubic, denoted  $\gamma$ ). There are no intermediate phases in the Mg-Mn system, see [2007Du] for the computed diagram.



Fig. 1 Al-Mn binary phase diagram in the Al-rich region [2006Pre]



Fig. 2 Al-Mn computed binary phase diagram [2007Du, 2009Shu]

l+bcc

895

1002

L+Al<sub>1</sub>Mn<sub>4</sub>(HT)

910

923



Fig. 3 Al-Mg-Mn computed liquidus projection [2007Du]



Fig. 4 Al-Mg-Mn computed liquidus projection near Mg corner [2007Du]

# **Ternary Thermodynamic Assessments**

[2007Du] accepted the binary descriptions of Al-Mg and Mg-Mn systems from [1998Lia] and [2005Gro], respectively, and used their version of the Al-Mn phase diagram. In the ternary optimization, only one ternary interaction parameter was found necessary for the liquid phase. The



Fig. 5 Al-Mg-Mn computed isothermal section at 710 °C [2007Du]



Fig. 6 Al-Mg-Mn computed isothermal section at 450 °C for Al-rich alloys [2007Du]. Narrow two-phase regions are omitted

CALPHAD-optimized parameters for the ternary compound  $Mn_2Mg_3Al_{18}(\tau)$  were in reasonable agreement with those derived from first-principles methodology [2007Du].

[2009Shu] carried out another thermodynamic assessment of this ternary system, using their assessed Al-Mn phase diagram with the binary descriptions of Al-Mg and Mg-Mn systems from the literature. Selected phase diagram and thermodynamic data were used in the optimization. The ternary

#### Section II: Phase Diagram Evaluations



Fig. 7 Al-Mg-Mn computed isothermal section at 400 °C for Al-rich alloys [2007Du]. Narrow two-phase regions are omitted



Fig. 8 Al-Mg-Mn computed vertical section at 5.05 mass% Al [2007Du, 2009Shu]

phase relationships could be reproduced satisfactorily, with one small additional ternary parameter for the liquid phase.

## **Computed Phase Equilibria**

The liquidus projection computed by [2007Du] is shown in Fig. 3. The details of the phase equilibria near the Mg



Fig. 9 Al-Mg-Mn computed solubility of Mn and Al in Mg-rich liquid at 730 °C [2007Du, 2009Shu]



Fig. 10 Al-Mg-Mn computed solubility of Mn and Al in Mg-rich liquid at 670 °C [2007Du, 2009Shu]

corner are shown in an enlarged view in Fig. 3, as well as in Fig. 4. Invariant reactions marked  $M_1$  and  $M_2$  by [2007Du] are transition reactions and are relabeled as  $U_0$  and U in Fig. 3. Reactions involving  $Al_3Mg_2$  and  $Al_{12}Mg_{17}$  are too close to the Al-Mg side and are not shown in Fig. 3. Phase equilibria close to the Al corner are omitted, as the mode of formation of the  $\lambda$  phase is not established.

Three isothermal sections at 710, 450 and 400 °C from the computed results of [2007Du] are shown in Fig. 5-7. The isothermal sections at 450 and 400 °C are for Al-rich





Fig. 11 Al-Mg-Mn computed solubility of Mg and Mn in solid Al [2007Du, 2009Shu]

alloys and are compared with the experimental data from the literature. [2009Shu] also computed a liquidus projection and isothermal sections at 1200, 850, 700, 450 and 400 °C.

The computed vertical sections at 5.05 mass% Al from the results of [2007Du] and [2009Shu] are compared in Fig. 8. The agreement between the two computed phase boundaries and with the experimental data is good. Figures 9 and 10 show the solubility of Mn and Al in (Mg)-rich liquid at 730 and 670 °C from the computed results of [2007Du] and [2009Shu]. The results of [2007Du] show better agreement with experimental data at 670 °C. Figure 11 shows the solubility of Mg and Mn in solid (Al) at 600, 550 and 500 °C as computed by [2007Du] and [2009Shu]. The results of [2009Shu] agree better with the experimental results.

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